Pressure effects on the charge-ordering transition of BaYCo₂O_{5,0}

Y. Moritomo, ^{1,2} M. Hanawa, ^{1,2} Sheng Xu, ³ H. Ishikawa, ¹ Y. Ohishi, ⁴ K. Kato, ⁴ T. Honma, ⁵ P. Karen, ⁶ M. Karppinen, ⁵ and H. Yamauchi ⁵

**Department of Applied Physics, Nagoya University, Nagoya 464-8603, Japan

Department of Applied Physics, Nagoya University, Nagoya 464-8603, Japan ²PRESTO, JST

³Department of Crystalline Materials Science, Nagoya University, Nagoya 464-8603, Japan
 ⁴JASRI/SPring-8, 1-1-1 Kouto, Mikazuki-cho, Sayo-gun, Hyogo 679-5198, Japan
 ⁵Materials and Structures Laboratory, Tokyo Institute of Technology, Yokohama 226-8503, Japan
 ⁶Department of Chemistry, University of Oslo, Blindern N-0315, Oslo, Norway
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Pressure effects on the charge-ordering (CO) transition have been investigated for an oxygen-deficient double-perovskite cobalt oxide, $BaYCo_2O_{5.0}$, through resistivity measurements using a piston-cylinder-type clamp cell. The charge-ordering temperature T_{CO} rises (+19 K/GPa) as pressure increases, making a sharp contrast with the pressure suppression of T_{CO} seen for the isostructural $BaSmFe_2O_{5.0}$. We have further determined the atomic coordinates under elevated pressures, and ascribed this unexpected behavior of $BaYCo_2O_{5.0}$ to the pressure enhancement of the orthorhombic distortion $s (\equiv a - b)$.

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I. INTRODUCTION

In perovskite-like 3d transition-metal oxides, the barely mobile d electrons together with the on-site Coulomb repulsion (U) cause a variety of charge-ordering (CO) and orbitalordering (OO) transitions. ¹⁻⁴ These CO and OO transitions are amenable to the structural modification via application of hydrostatic pressure (physical pressure) and/or isovalent chemical substitution (chemical pressure),⁵ because the charge and orbital degree of freedoms couple with the shape of the oxygen polyhedron surrounding the transition metal. In this sense, structural studies are effective for deeper understanding of CO and OO phenomena. Synchrotron x-ray powder diffraction together with the diamond anvil cell (DAC) technique enables us to determine oxygen positions even under high pressures.^{6,7} Then, we can determine the structural modification under physical pressures, and clarify the physical pressure effects on the CO and OO phenomena.

Among the CO compounds, the double-perovskite oxides of the $BaRCo_2O_5$ type (R = Y, Ho, Dy, and Tb), attract considerable scientific interest.^{2,3,8} BaYCo₂O₅ has an orthorhombic crystal structure (Pmmm; Z=2), containing corner-shared CoO₅ square pyramids as principal building units, and is isostructural with BaSmFe2O5 that shows a "Verwey-type" transition at $T_{\rm CO} = 230$ K. BaYCo₂O₅ undergoes an antiferromagnetic transition at 330 K: the spin ordering (SO) is of the checker-board type (G type) within the bipyramidal layer. With further decrease of temperature below $T_{\rm CO}$ (\approx 220 K), chain-type CO along the b axis takes place, accompanied by a kink in resistivity.² Here, we emphasize that the CO and SO patterns of BaYCo₂O₅ are the same as those of BaSmFe₂O₅. Having similar T_{CO} , the two compounds make a good couple for comparison. Akahoshi and Ueda⁴ insisted that BaYCo₂O₅ separates into orthorhombic and tetragonal phases below $T_{\rm CO}$, and that there is no anomaly in resistivity at $T_{\rm CO}$ though there is a clear anomaly in differential scanning calorimetry (DSC) curve. They ascribed the discrepancy of the data between Ref. 2 and Ref. 4 to the different oxygen stoichiometry.

In this paper, we report the physical pressure effects on $T_{\rm CO}$ and on the crystal structure for BaYCo₂O₅. We have found that $T_{\rm CO}$ of BaYCo₂O₅ rises (+19 K/GPa) as pressure increases, making a sharp contrast to the pressure suppression of $T_{\rm CO}$ in BaSmFe₂O₅. This unexpected behavior is ascribed to the pressure enhancement of the orthorhombic distortion s ($\equiv a-b$) in BaYCo₂O₅. Actually, $T_{\rm CO}$ under physical and chemical pressures scales well to s for both BaYCo₂O₅ and BaSmFe₂O₅.

II. EXPERIMENT

Polycrystalline sample of BaYCo₂O_{5+ δ} with $\delta \approx 0$ was obtained through a two-step synthesis route. The direct solidstate synthesis in air starting from a stoichiometric mixture of BaCO₃, Y₂O₃, and Co₃O₄ readily results in single-phase BaYCo₂O_{5+ δ} with δ >0. Here, calcination was carried out at 1000 °C for 10 h and sintering of a pelletized sample at 1100 °C for 24 h. For the thus-obtained air-synthesized sample, iodometric titration yielded the amount of excess oxygen at $\delta = 0.32(1)$. The excess oxygen was then completely removed by annealing in flowing N₂ gas (in a thermobalance) at 800 °C for 12 h. For this sample δ was determined to be 0.005(3). Differential scanning calorimetry runs showed for the sample a clear peak in both heating and cooling curves around 231-233 K, the precise temperature depending on the heating/cooling rate. Synthesis procedure of the BaSmFe₂O_{5.0} sample was described elsewhere.^{9,10}

The resistivity (ρ) was measured under pressures up to P=1.6 GPa by a standard four-probe method with the use of a clamp-type piston cylinder cell.⁵ A small piece of the polycrystalline sample ($\sim 0.5 \times 1 \times 2 \text{ mm}^3$) was placed in a sample room, which was filled with Silicone oil as a pressure-transmitting medium. The sample temperature was monitored with a copper-constantan thermocouple attached near the bottom of the sample space. The pressure values are

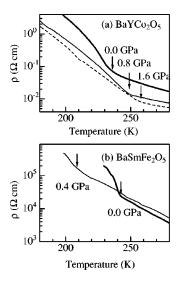


FIG. 1. Physical pressure effects on resistivity ρ of (a) BaYCo₂O_{5,0} and (b) BaSmFe₂O_{5,0}. The pressure values quoted are those measured at room temperature. Downward arrows indicate the charge-ordering temperatures $T_{\rm CO}$.

those measured at room temperature. (The applied pressure relaxes at a rate of -7%/100 K.) Pressure-induced changes in the ρ -T curves were reproducible in the repeated pressure cycles.

High-pressure x-ray powder-diffraction measurements were performed at SPring-8, BL10XU beamline equipped with the DAC in the 2θ range of $5^{\circ}-30^{\circ}$. Precipitation method¹¹ was adopted to obtain fine and homogeneous powders: the waiting time was 120 min (30 min) for BaYCo₂O₅₀ (BaSmFe₂O_{5.0}). The sample powder was sealed in the DAC gasket hole, 0.05 mm in thickness and 0.15 mm in diameter, filled with an ethanol/methanol mixture as a pressuretransmitting medium. The x-ray beam size was 0.10 mm in diameter. The wavelength of the incident x ray was 0.495 59 Å, and the exposure time was 8 min. The Debye-Scherrer rings gave homogeneous intensity distributions, which is a necessary condition for successful Rietveld refinement. The magnitude of the applied pressure was monitored by the wavelength shift of the luminescence line R_1 (Ref. 12) from a small piece of ruby placed in the gasket hole.

III. RESULTS AND DISCUSSION

A. Pressure effects on the charge-ordering temperature

In Fig. 1(a) temperature dependence of ρ is shown for BaYCo₂O_{5.0} under various pressures. The ρ -T curve at ambient pressure (P=0.0 GPa; thick curve) shows a steep rise at 234 K for BaYCo₂O_{5.0}, due to the carrier localization effect in the charge-ordered phase. The charge-ordering temperature $T_{\rm CO}$ (=234 K) at ambient pressure is close to the transition temperature (=231–233 K) observed in the DSC measurement. $T_{\rm CO}$ gradually rises as pressure increases. These $T_{\rm CO}$ values are plotted as open circles in Fig. 2 against the applied pressure. The initial pressure derivative of the charge-ordering temperature ($dT_{\rm CO}/dP$) is +19 K/GPa. The isostructural BaSmFe₂O_{5.0} also shows the charge-

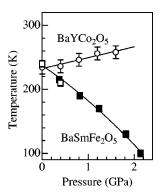


FIG. 2. Pressure dependence of the charge-ordering temperature $T_{\rm CO}$. Open circles and squares represent data determined in this study for BaYCo₂O_{5.0} and BaSmFe₂O_{5.0}, respectively. Closed squares are data points for BaSmFe₂O_{5.0} from Ref. 9. The straight line for BaYCo₂O_{5.0} is least-squares-fitted. The curved line for BaSmFe₂O_{5.0} is a guide for the eye.

ordering transition at $T_{\rm CO}$ =240 K, as indicated by arrows in Fig. 1(b). In contrast to BaYCo₂O_{5.0}, $T_{\rm CO}$ of BaSmFe₂O_{5.0} steeply decreases to \approx 205 K at P=0.4 GPa. [Unfortunately, we cannot determine $T_{\rm CO}$ above P>0.4 GPa because ρ becomes too high (>10⁵ Ω cm) to measure.] These $T_{\rm CO}$ values are plotted as open squares in Fig. 2 against the applied pressure.

B. Temperature effects on crystal structure

Figure 3(a) shows temperature variation of the lattice constants a, b and c and unit cell volume (V, closed circles) for BaYCo₂O_{5.0} in the Pmmm description. The lattice constants were refined by the Rietveld analysis [RIETAN2000 (Ref. 13)] on the x-ray powder diffraction data obtained at SPring-8, BL02B2 beam line¹⁴ equipped with a large Debye-Scherrer camera.¹⁵ As temperature decreases below $T_{\rm CO}$, the orthorhombic distortion s ($\equiv a-b$) increases. Here, we em-

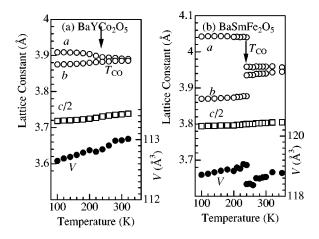


FIG. 3. Temperature dependence of lattice constants a, b, and c and unit cell volume (V, closed circles) of (a) BaYCo₂O_{5.0} and (b) BaSmFe₂O_{5.0}. The data of the latter compound are from Ref. 9. Downward arrows show the charge-ordering temperature $T_{\rm CO}$. Pmmm~(Z=2) description was adopted also below $T_{\rm CO}$ for convenience.

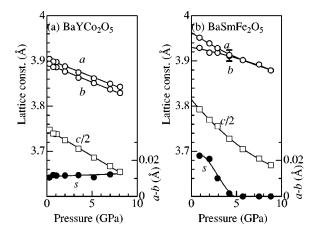


FIG. 4. Pressure dependence of lattice constants a, b and c and orthorhombic distortion s ($\equiv a-b$: closed circles) of (a) BaYCo₂O_{5,0} and (b) BaSmFe₂O_{5,0} at 300 K.

phasize that the unit cell volume V shows a slight increase at $T_{\rm CO}$, which rules out the simple volume effects as the origin for the pressure enhancement of $T_{\rm CO}$ in BaYCo₂O_{5.0}. Note that the magnitude of the increase in V upon cooling through $T_{\rm CO}$ agrees well with that previously reported for BaYCo₂O_{5.0}.

Figure 3(b) shows temperature variation of a, b, c, and V for isostructural BaSmFe₂O_{5.0}. Similarly to the case of BaYCo₂O_{5.0}, both the s and V values increase as temperature decreases below $T_{\rm CO}$. For V the increase is as large as 0.7% upon cooling through $T_{\rm CO}$. This fact dictates that physical pressure will suppress $T_{\rm CO}$ of BaSmFe₂O_{5.0} by decreasing the stability of the low-temperature phase that has larger volume.

C. Pressure effects on structure

Figure 4(a) shows pressure effects on the lattice constants a, b and c determined from Rietveld refinements and on the orthorhombic distortion s ($\equiv a-b$) for BaYCo₂O_{5.0} at 300 K. In line with the pseudotetragonal ($a \approx b \geq c/2$) structure, the pressure effect is found to be anisotropic. The pressure coefficient of c [$\equiv -1/c(dc/dP) = 0.31\%$ /GPa) is larger than those of the other two lattice constants, a (0.21%/GPa) and b (0.20%/GPa). We point out that the s value slightly increases with pressure. Figure 4(b) shows pressure effects on a, b, c, and s for BaSmFe₂O_{5.0} at 300 K. The data of BaSmFe₂O_{5.0} above P > 5 GPa were analyzed with the tetragonal model (a = b). The magnitude of s for BaSmFe₂O_{5.0} steeply decreases with pressure and disappears above P = 4 GPa, making sharp contrast with the behavior seen for BaYCo₂O_{5.0}.

Here, we point out that the pressure effect on s correlates with that on $T_{\rm CO}$ (compare Fig. 2 and Fig. 4). We plotted in Fig. 5 $T_{\rm CO}$ (open circles) of the compressed BaYCo₂O_{5.0} against s. The filled circles represent the data for isostructural phases BaHoCo₂O₅ and BaTbCo₂O₅. By using the quantity s as a scaling parameter, both the physical and the chemical pressure data are found to form a single unified curve. This suggests that the orthorhombic distortion s is significance for

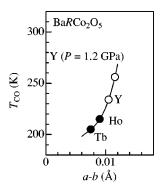


FIG. 5. Interrelation between $T_{\rm CO}$ and orthorhombic distortion s ($\equiv a-b$) for BaYCo₂O₅. Filled circles represent BaHoCo₂O₅ (Ref. 8) and BaTbCo₂O₅ (Ref. 8). The curve is a guide for the eye. The error bars for the horizontal axis are within the symbol size.

the stability of the CO state in BaRCo₂O₅. Here, we note that a similar scaling relation between s and T_{CO} is observed also in BaRFe₂O_{5.0}. This common feature suggests that the correlation originates in the chain-type CO along the b axis, for which the orthorhombic distortion is indispensable. ^{16,17} Fulfillment of this correlation between s and T_{CO} may take priority over the volume change effect, and we see that despite a slight volume increase upon the CO transition in BaYCo₂O_{5.0}, the applied pressure increases T_{CO} .

D. Pressure effects on atomic coordinates

Finally, let us investigate deformation of the MO_5 square-pyramidal shape under physical pressure. Figure 6 shows pressure effects on the atomic coordinates of (a) BaYCo₂O_{5.0} and (b) BaSmFe₂O_{5.0} at 300 K. For the Rietveld structural refinement, the oxygen z coordinates at the 2s and 2r Wyckoff sites (Pmmm) were assumed to be the same (z_0) and the overall isotropic atomic displacement parameter Q was adopted. In BaYCo₂O_{5.0}, the reliability factor $R_1 (\equiv \sum_K |I_{K,\text{calc}} - I_K|/\sum_K I_{K,\text{calc}})$ was small enough (1.80% to 2.51%). R_1 of BaSmFe₂O_{5.0}, however, is rather large (1.76% to 4.72%) due to the slight broadening of the diffraction profiles, especially in the high-pressure region. Except for

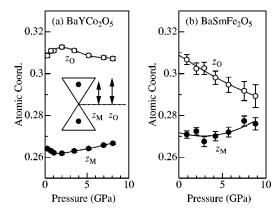


FIG. 6. Pressure dependence of atomic coordinate of (a) $BaYCo_2O_{5,0}$ and (b) $BaSmFe_2O_{5,0}$ at 300 K. The origin of the *z* coordinate is at the level of apical oxygens, as shown in the inset. Solid curves are guides to the eye.

the lower-pressure region of BaYCo₂O_{5.0}, physical pressure decreases the z coordinate of the base oxygen atom of the square pyramid [$z_{\rm O}$; see Fig. 6(a)], and hence compresses the pyramid along the c axis. The analogous compression is much enhanced for BaSmFe₂O_{5.0}. This is because the SmO bond distance ($d_{\rm SmO}$ = 2.45 Å at 0.9 GPa) of BaSmFe₂O_{5.0} is already comparable with the sum of the ionic radii [= 2.46 Å (Ref. 18)] of O²⁻ and Sm³⁺, and hence the reduction of c pushes down the oxygen coordinate toward the oxygen apex. Actually, the SmO bond distance is nearly unchanged even under pressures: $d_{\rm SmO}$ = 2.46 Å (2.47 Å) at 2.9 GPa (5.7 GPa).

IV. SUMMARY

In summary, we have investigated pressure effects on the charge-ordering transition in BaYCo₂O_{5.0}. Surprisingly, the charge-ordering temperature $T_{\rm CO}$ rises (+19 K/GPa) as pressure increases, making a sharp contrast with the pressure

suppression of $T_{\rm CO}$ in the isostructural BaSmFe₂O_{5.0}. This unexpected behavior of BaYCo₂O_{5.0} is ascribed to the pressure enhancement of the orthorhombic distortion s. We have further found that $T_{\rm CO}$ scales to s under both physical and chemical pressures. Thus, high-pressure structural approach was demonstrated to be a powerful tool to understand the charge-ordering phenomena in transition metal oxides.

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¹⁵The x-ray diffraction patterns of our BaYCo₂O_{5.0} powders can be reproduced by single phase model with *Pmmm* structure at least down to 100 K. The reliable factor $R_{\rm I}$ (≡[$\Sigma_i w_i(y_i - y_{i, {\rm calc}})^2/\Sigma_i w_i y_i^2$]^{1/2}) is 4.0–4.4 % below 150 K.

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